

# Co<sub>2</sub>CrIn: A Further Magnetic Heusler Compound

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*Dedicated to Professor Wolfgang Jeitschko on the occasion of his 70<sup>th</sup> birthday*

A further example of the class of Heusler compounds is presented. Co<sub>2</sub>CrIn is L2<sub>1</sub> ordered (face centered cubic, space group  $Fm\bar{3}m$ ) with a lattice constant of  $a = 6.0596(2)$  Å. The crystal structure was determined from powder diffraction data by means of the Rietveld method. The magnetic properties of Co<sub>2</sub>CrIn were measured by means of SQUID magnetometry. The material turns out to be a soft ferromagnet with a saturation moment of  $1.2 \mu_B$  at 5 K.

**Key words:** Magnetic Properties, Heusler Compounds

## Introduction

The Co<sub>2</sub>YZ Heusler compounds (with  $Y$  = transition metal and  $Z$  = main group metal) exhibit the highest Curie temperatures (1100 K, Co<sub>2</sub>FeSi [1–3] and the highest magnetic moments ( $6.0 \mu_B$  per unit cell, Co<sub>2</sub>FeSi [1–3]) being reported up to now.

In particular, some Co<sub>2</sub>CrZ Heusler compounds with  $Z$  = (Al, Ga) are reported to be half-metallic ferromagnets [4–8]. The magnetic moment of Co<sub>2</sub>CrGa is reported by Umetsu *et al.* [9] to be  $3.01 \mu_B$  [9]. This value is in accordance with the Slater-Pauling rule. In particular, the measured magnetic moment is an integer value, which points to a half-metallic character for Co<sub>2</sub>CrGa. Co<sub>2</sub>CrAl is also predicted to be a half-metallic ferromagnet. In experiments, however the compound reveals a magnetic moment of only  $1.55 \mu_B$  [10]. This value is half of the value obtained from the Slater-Pauling rule [11–13] and is no integer value.

The quaternary Heusler compounds Co<sub>2</sub>Cr<sub>1-x</sub>Fe<sub>x</sub>Al show also magnetic moments, which are significantly lower in bulk materials [4–8, 10, 14, 15] and in tunnel junctions [16–19] compared to the value [11–13] expected from the Slater-Pauling rule.

This reduction of the magnetic moment might originate from an interchange of Cr and Co sites, as proposed by Miura *et al.* [6, 7]. This type of disorder leads to an antiferromagnetic coupling of the antisite-Cr with the nearest neighbour ordinary site-Cr atoms and de-

grades the half-metallic character. Zhang *et al.* [15] estimated the Co-(Cr,Fe) (DO<sub>3</sub> type) disorder to be about 30%. X-ray magnetic circular dichroism measurements of Co<sub>2</sub>Cr<sub>1-x</sub>Fe<sub>x</sub>Al revealed that the measured site specific moment of the Cr is significantly smaller than the calculated value. This points to disorder effects and selective oxidation of Cr [20, 21]. The site specific results are in agreement with measurements of the total moments as mentioned above.

Nevertheless, pressed powder samples of Co<sub>2</sub>Cr<sub>0.6</sub>-Fe<sub>0.4</sub>Al show a very high magnetoresistance ratio (30%) at room temperature [4, 5] in bulk material and large tunneling magnetoresistance between 19% [16–18] and 42% [19, 22] in tunnel junctions. The equi-electronic Co<sub>2</sub>Cr<sub>1-x</sub>Fe<sub>x</sub>Ga compounds crystallise in the ordered L2<sub>1</sub> structure and the magnetic moments for the compounds with low concentration of  $x$  are in good agreement with the Slater Pauling rule [23, 24].

A comprehensive characterisation of Cr containing Heusler compounds is necessary in order to understand the mechanism of the reduced magnetic moment of Cr in the Co<sub>2</sub>Cr<sub>1-x</sub>Fe<sub>x</sub>Al compounds in contrast to the Co<sub>2</sub>Cr<sub>1-x</sub>Fe<sub>x</sub>Ga compounds. While Co<sub>2</sub>YZ Heusler compounds with  $Z$  = Al, Ga are already known, the next compound with an element of the series of main group elements is a Co<sub>2</sub>YZ Heusler compound with  $Z$  = In. This compound has not been reported up to now. Therefore (see Fig. 1) Co<sub>2</sub>CrIn was synthesised and characterised in the present study.

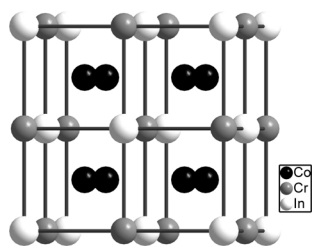


Fig. 1. Crystal structure of Co<sub>2</sub>CrIn.

## Experimental Section

Polycrystalline samples of Co<sub>2</sub>CrIn were prepared by repeated arc-melting of stoichiometric quantities of the pure metals in a purified argon atmosphere. The weight loss was monitored to be less than 0.5%.

The Heusler type (L<sub>21</sub>) structure was determined by X-Ray powder diffraction (Bruker AXS D8) with a mixture of Cu-K<sub>α1</sub> and Cu-K<sub>α2</sub>-radiation. The measurements were performed in reflection geometry in a  $\theta$ - $\theta$  scanning mode. A multilayer Göbel mirror was used as monochromator. The  $2\theta$  scan range was set from 15° to 115.106° with a step size of 0.0357° leading to 2804 independent measurement points. All measurements were performed at room temperature. The structural parameters of Co<sub>2</sub>CrIn were obtained from refinement of the XRD data using the Jana2000 program package [25].

The magnetic properties were investigated at low temperatures using a superconducting quantum interference device (SQUID, Quantum Design MPMS-XL-5).

## Results and Discussion

Heusler compounds [26] are ternary intermetallics with stoichiometric composition  $X_2YZ$ . Fig. 2 shows the diffractogram of Co<sub>2</sub>CrIn. The data was refined in the face centered cubic Heusler type L<sub>21</sub> structure (BiF<sub>3</sub> type) [27]. A scheme of the Heusler type structure can be seen in Fig. 1. All reflections of Co<sub>2</sub>CrIn were identified using the space group  $Fm\bar{3}m$ , number 225. The Co atoms are located on the 8c Wyckoff position ( $1/4, 1/4, 1/4$ ), the Cr atoms on 4b ( $1/2, 1/2, 1/2$ ) and In on 4a (0 0 0). There are 4 formula units per unit cell leading to an overall composition of Co<sub>8</sub>Cr<sub>4</sub>In<sub>4</sub>. The background was refined using 9 terms of a Legendre Polynom and the  $2\theta$ -range of 15°–22° was excluded from the refinement. Within the fitting procedure, the peak form was assumed to have a Lorentz-shape with  $L_x = 8.2(5)$  and  $L_y = 21.5(13)$ .

The lattice constant is determined with this refinement parameter to be 6.0596(2) Å. The positional and atomic displacement factors  $U_{iso}$  are small with  $U_{iso}$  (Co) = 0.019(2),  $U_{iso}$  (Cr) = 0.027(2) and  $U_{iso}$  (In) =

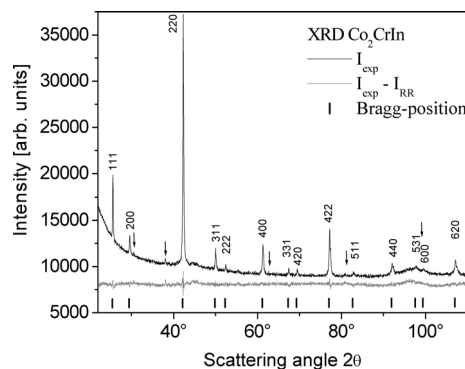


Fig. 2. Diffraction pattern and difference plot of the Rietveld refinement of Co<sub>2</sub>CrIn. Shown are the measured intensity ( $I_{exp}$ ) and the difference compared to the Rietveld refinement ( $I_{RR}$ ). Vertical bars indicate the Bragg positions of the expected diffraction peaks. Reflections of a secondary phase, which could not be identified, are marked with arrows.

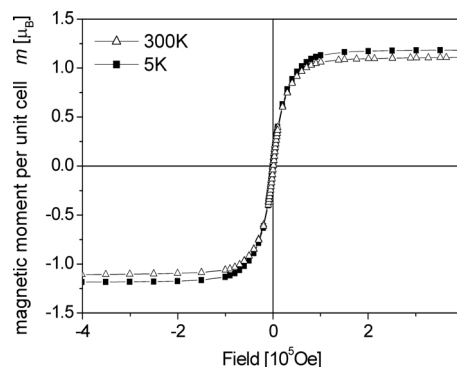


Fig. 3. Magnetic hysteresis curve of Co<sub>2</sub>CrIn at 5 K and 300 K, respectively.

0.007(2). The cell volume is 222.506(8) Å<sup>3</sup> and the calculated density amounts to 8.495(3) g/cm<sup>3</sup>.

The quality of the fit was denoted in the values of  $R = 3.59$ ,  $R_{wobs} = 1.5$  and  $R_{wall} = 1.54$ . In addition, the diffractogram of Co<sub>2</sub>CrIn exhibits additional reflections due to a small amount of an unidentified phase. Further details of the crystal structure investigation are available from the Fachinformationszentrum Karlsruhe, D-76344 Eggenstein-Leopoldshafen (Germany), depository number CSD-416260.

The refinement of Co<sub>2</sub>CrIn revealed no large disorder effects in contrast to Co<sub>2</sub>CrAl. A high degree of disorder arising from the Co and Cr positions (DO<sub>3</sub> disorder) could be excluded from the refinement. Any disorder influences the intensities of the (111) and (200) reflection. Thus the ratio of the intensities of the (111) and (200) reflection might be used as an indica-

tor of disorder effects in this class of compounds. In Co<sub>2</sub>CrIn, the cross sections of Cr and In are of different orders of magnitude in contrast to Cr and Ga. Thus the ratio of the intensities of the (111) and (200) reflections of Co<sub>2</sub>CrIn is more significant in Co<sub>2</sub>CrIn than in Co<sub>2</sub>CrGa and Co<sub>2</sub>CrAl and disorder effects are more apparent.

The magnetic properties were measured by means of SQUID magnetometry. Fig. 3 shows the hysteresis curves of Co<sub>2</sub>CrIn measured at 5 K and 300 K. The magnetic moment in saturation is 1.18  $\mu_B$  at 5 K and 1.1  $\mu_B$  at 300 K corresponding to a magnetic moment of 0.3  $\mu_B$  per atom at 5 K and 0.28  $\mu_B$  per atom at 300 K, respectively. Co<sub>2</sub>CrIn turns out to be a soft ferromagnet with a small remanence and a small coercive field.

Many Co<sub>2</sub>YZ Heusler compounds follow the Slater-Pauling curve [11,12], as described by Kübler [13]. An application of this *valence electron rule* on Co<sub>2</sub>CrIn leads to a magnetic moment of 3  $\mu_B$ . This is the expected magnetic moment for all Co<sub>2</sub>CrZ compound with Z being a third main group element, if the compound is predicted to be a half-metallic ferromagnet. Thus the measurement reveals a reduced magnetic mo-

ment and excludes half-metallic ferromagnetic ordering. The structural refinement showed no major disorder. Therefore the reduced magnetic moment is not due to disorder effects and may arise from a ferri – or an antiferromagnetic coupling of the Co and Cr atoms.

## Summary and Outlook

In summary, this work presents the synthesis and characterisation of the Heusler compound Co<sub>2</sub>CrIn. The compound is L2<sub>1</sub> ordered and shows no major antisite disorder. Co<sub>2</sub>CrIn turns out to be a ferrimagnet with a magnetic moment of 1.18  $\mu_B$  at 5 K. In addition, the hysteresis curve reveals a soft magnetic behaviour. A measurement of the site specific magnetic moments is highly desirable as it might lead to a deeper understanding of the magnetic properties of Co<sub>2</sub>CrIn. Furthermore, the measured magnetic moment is not an integer number, as expected for a half-metallic ferromagnet. Thus, Co<sub>2</sub>CrIn can not be a half-metallic ferromagnet, as most of the other Co<sub>2</sub>YZ Heusler compounds.

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